

## Preparation and Characterization of (TiO<sub>2</sub>-SnO<sub>2</sub>) Thin Films by Pulsed Laser Deposition

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### ABSTRACT

In this work, mixed oxide (TiO<sub>2</sub>-SnO<sub>2</sub>) thin films were grown on Si (111) and glass substrates by pulsed laser deposition (PLD) method. The influences of increasing amounts of SnO<sub>2</sub> were investigated. The X-ray diffraction results show the peaks position of the plane was shifted towards higher angle values with increasing amounts of SnO<sub>2</sub>. The surface morphology of the deposits materials was also studied by using a scanning electron microscope (SEM). The results show that, the grain sizes decrease with increasing SnO<sub>2</sub> content from the largest value (53.6) nm to smallest value (25.5) nm. From UV-visible spectroscopy, the distinct variations in the transmission spectra, and optical energy gap, of the thin films were also observed.

**Keywords:** Pulsed Laser Deposition (PLD), (TiO<sub>2</sub>-SnO<sub>2</sub>) Thin Films, Structural Properties, Surface Morphology, Optical Properties

### تحضير ودراسة اغشية رقيقة مكونة من خليط اوكسيد التيتانيوم و اوكسيد القصدير بطريقة الترسيب بالليزر النبضي

#### الخلاصة

يتضمن هذا العمل ترسيب اغشية رقيقة من خليط اوكسيد التيتانيوم و اوكسيد القصدير على السيلكون والزجاج باستخدام طريقة الترسيب بالليزر النبضي وقد تم مناقشة تأثير زيادة نسبة اوكسيد القصدير على خصائص خليط اوكسيد التيتانيوم و القصدير. وقد بينت نتائج حيود الاشعة السينية ان موقع القمم انحرفت باتجاه قيم الزوايا الاعلى بزيادة نسبة اوكسيد القصدير. اما مورفولوجية السطح للمادة المترسبة فقد تم دراستها بواسطة المجهر الماسح الالكتروني. وقد بينت النتائج ان الحجم الحبيبي للجسيمات النانوية قل بزيادة نسبة اوكسيد القصدير من الحجم الحبيبي (53.6 نانومتر) الى الحجم الحبيبي (25.5 نانومتر). وكذلك قد تم ملاحظة التغيرات الحاصلة لطيف النفاذية وكذلك فجوة الطاقة البصرية بواسطة قياسات مطياف النفاذية للأشعة المرئية وفوق البنفسجية.

## INTRODUCTION

Mixed oxide systems have attracted considerable attention. The good stability of sensing properties of SnO<sub>2</sub> for reducing gases, combined with the good chemical stability of TiO<sub>2</sub> at high temperatures, stimulate the study on the applications of TiO<sub>2</sub>-SnO<sub>2</sub>. In particular, TiO<sub>2</sub>-SnO<sub>2</sub> system combines the positive features of both materials being used in gas detection [1] and suggested to be applied as high-temperature resistors.[2].

It was concluded by K. Zakrzewska and co-workers that high operating temperatures of TiO<sub>2</sub> sensors could be reduced to about 770 K as a result of Sn incorporation. [3].

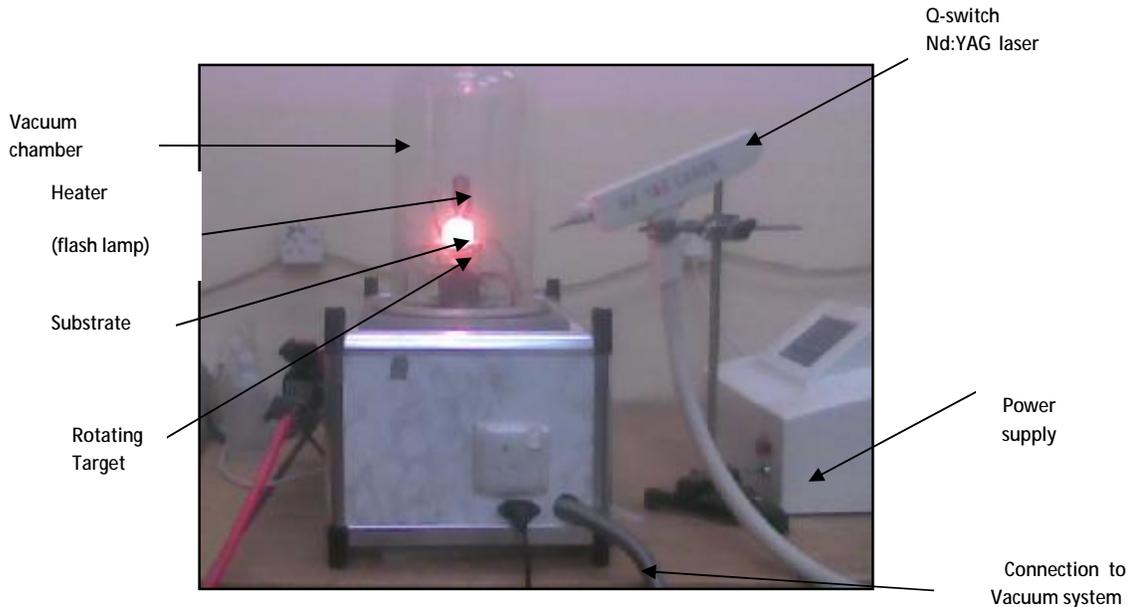
There are many different techniques used for depositing tin oxide films: r. f. sputtering, dc-magnetron sputtering, thermal evaporation, ion beam deposition, rheotaxial growth and thermal oxidation (RGTO), chemical vapour deposition, spray pyrolysis, successive ionic layer deposition (SILD) and other chemical methods. Sberveglieri has presented a review of the techniques applied for oxide films deposition[4],[5]., all methods discussed require high substrate temperature or post deposition annealing in order to fabricate good quality polycrystalline films. High temperature, however, damages the surface of the films and increases the interface thickness, which has negative effect on the optical properties, especially on the wave guiding. Pulsed laser deposition technique was successfully applied for growing of quality thin films [6].

This technique is also suitable for depositing oxide films at a relative high deposition rate and low cost [7, 8]. In this work, we report on the growth of (TiO<sub>2</sub>-SnO<sub>2</sub>) deposits by PLD using 10 ns pulses at 532 nm on Si (111) and glass substrates. The deposits were characterized by X-ray diffraction (XRD) to examine their crystallinity, scanning electron microscope (SEM) to observe the surface structure and UV-visible spectroscopy to investigate the optical properties of the films.

## EXPERIMENTAL PROCEDURE

### Film preparation

The deposition was carried out using a Q-switched Nd:YAG laser with a second harmonic generation (SHG) at wavelength is 532nm with pulse width 7ns and repetition rate 10Hz. The studied films were prepared by mixed oxides (TiO<sub>2</sub>-SnO<sub>2</sub>) films with different SnO<sub>2</sub> contents (25%, 50% and 75%) targets. Films were grown by pulsed laser deposition on Si(111) and glass substrates kept an on-axis distance of 4cm from the target. The chamber was kept at vacuum pressure of 10<sup>-5</sup> mbar as shown in Figure (1). The (TiO<sub>2</sub>-SnO<sub>2</sub>) disc was ablated from 10-100 pulses (10-20 min) to get single layered thin films. Consequently, the films were deposited by PLD at 400 °C substrate temperature in an O<sub>2</sub> pressure 5×10<sup>-1</sup> mbar and laser fluence (1.4)J/cm<sup>2</sup>.



**Figure (I) Experimental setup**

### Film characterization

The crystalline structure of the films was determined by X-Ray Diffraction (XRD) measurements (Philips PW 1050,  $\lambda=1.54 \text{ \AA}$ ) using Cu  $k\alpha$ . Transmission measurements were performed for a range 300-800 nm using UV-VIS-PV-8800 (Perkin Elemer Company) spectrophotometer. The characterizations included determination of the absorption as a function of incident photon energy, determination of the transmission as a function of incident photon energy and determination the value energy gap. The surface morphology was examined by Scanning Electron Microscopy (SEM-JEOL 7000).

### RESULT AND DISCUSSION

Figure (2) shows the XRD patterns of the (TiO<sub>2</sub>-SnO<sub>2</sub>) films grown on Si (111) at  $T_s = 400 \text{ }^\circ\text{C}$  at laser fluence  $1.4 \text{ J/cm}^2$ . Diffraction peaks located at  $2\theta=28^\circ$  corresponding to Silicon substrates are shown in the Figure below. At SnO<sub>2</sub> 25% concentration, showed diffraction peaks located at  $2\theta=26.9^\circ$ ,  $2\theta=34^\circ$  and  $2\theta=52.1^\circ$  corresponding to the (110), (101) and (211), peaks respectively. At SnO<sub>2</sub> content to 50%, diffraction peaks were located at  $2\theta=27.0^\circ$ ,  $2\theta=34.3^\circ$  and  $2\theta=52.6^\circ$ , corresponding to the (110), (101) and (211) peaks respectively. At 75% concentration, showed diffraction peaks located at  $2\theta=27.25^\circ$ ,  $2\theta=34.5^\circ$  and  $2\theta=53^\circ$  corresponding to the (110), (101) and (211) where the peaks position of the plane was shifted towards higher angle ( $2\theta$ ) values with increasing amounts of SnO<sub>2</sub> content. It has been reported that TiO<sub>2</sub> crystallizes as anatase but even a small addition of Sn changes the crystallographic structure to that of tetragonal rutile. X-ray diffraction peaks are shifted from the positions characteristic to TiO<sub>2</sub>-rutile due to the change in the lattice parameters upon the substitution of Sn for Ti. With the increasing tin content, X-ray diffraction lines shift systematically towards positions

typical for tetragonal, cassiterite form of SnO<sub>2</sub>. All XRD patterns of films reflect that the crystal lattice of mixed oxide (SnO<sub>2</sub>-TiO<sub>2</sub>) did not undergo significant changes. XRD analysis also did not detect the TiO<sub>2</sub> phase, these due to the small molecular weight of TiO<sub>2</sub> as compared with that of SnO<sub>2</sub> and dispersion of this small-grain phase.

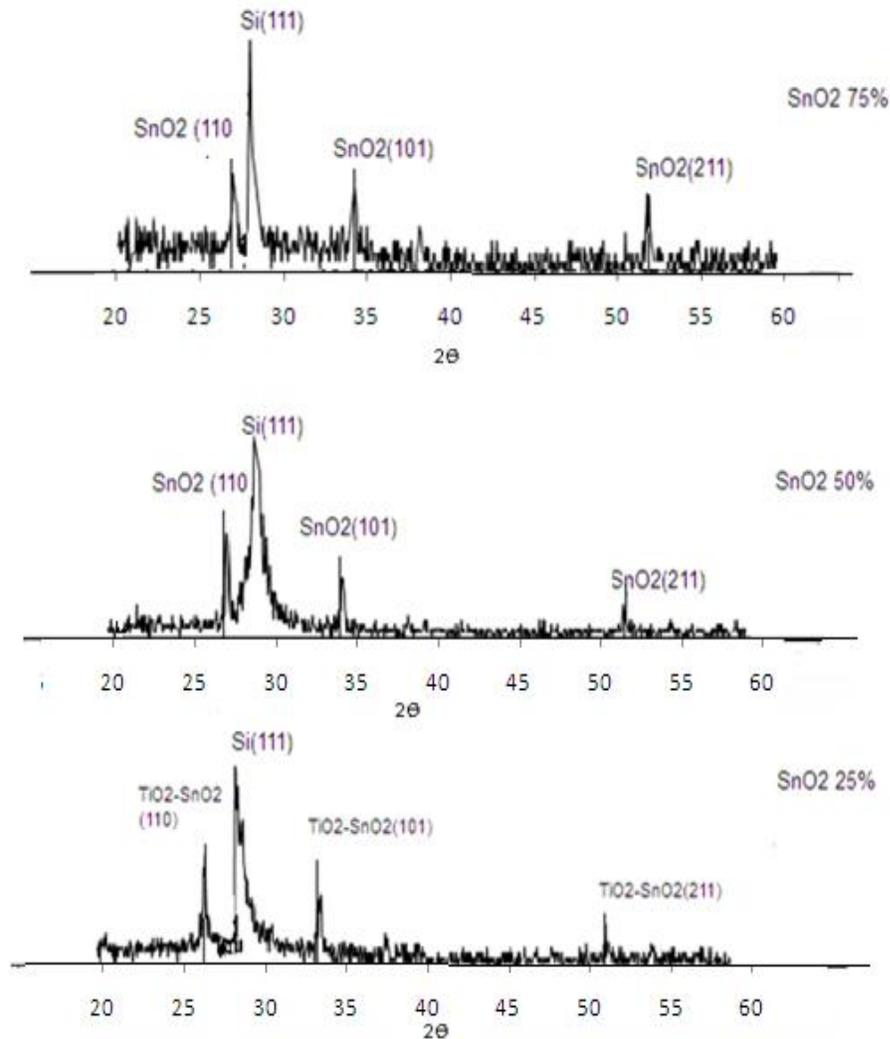


Figure (2) XRD patterns of (TiO<sub>2</sub> -SnO<sub>2</sub>) films grown on Si at various SnO<sub>2</sub> content.

**Table (1) the obtained result of the XRD for (TiO<sub>2</sub>-SnO<sub>2</sub>)/Si at T=400 °C.**

sample	2θ(degree)	(hkl)	FWHM <sup>0</sup>
(75%TiO <sub>2</sub> - 25%SnO <sub>2</sub> )	26.9	R(110)	0.163
	34	R(101)	0.164
	52.1	R(211)	0.143
	28	Si(111)	0.305
(50%TiO <sub>2</sub> - 50%SnO <sub>2</sub> )	27	R(110)	0.245
	34.3	R(101)	0.275
	52.6	R(211)	0.191
	28	Si(111)	0.305
(25%TiO <sub>2</sub> - 75%SnO <sub>2</sub> )	27.25	R(110)	0.1
	34.5	R(101)	0.144
	53	R(211)	0.157
	28	Si(111)	0.305

SEM images of the TiO<sub>2</sub> mixed with different content of SnO<sub>2</sub> (25%, 50%, 75%) are presented in Figure (3) for film deposited at fixed substrate temperature of 400 °C at Oxygen pressure of (5 × 10<sup>-1</sup> mbar) and 1.4 J/cm<sup>2</sup> laser fluence. The grain size decreases with increasing SnO<sub>2</sub> content. SEM images show clearly that size and shape of grains are strongly affected by the chemical composition of SnO<sub>2</sub>-TiO<sub>2</sub>. Grain growth has been observed for TiO<sub>2</sub>. Were Addition of SnO<sub>2</sub> reduces the grain size.

**Table (2) The obtained result of the SEM for  
(TiO<sub>2</sub>-SnO<sub>2</sub>)/Si at T=400 °C.**

sample	SEM of plane grain size (nm)
(75%TiO <sub>2</sub> - 25%SnO <sub>2</sub> )	53.6
(50%TiO <sub>2</sub> - 50%SnO <sub>2</sub> )	46.2
(25%TiO <sub>2</sub> - 75%SnO <sub>2</sub> )	25.5

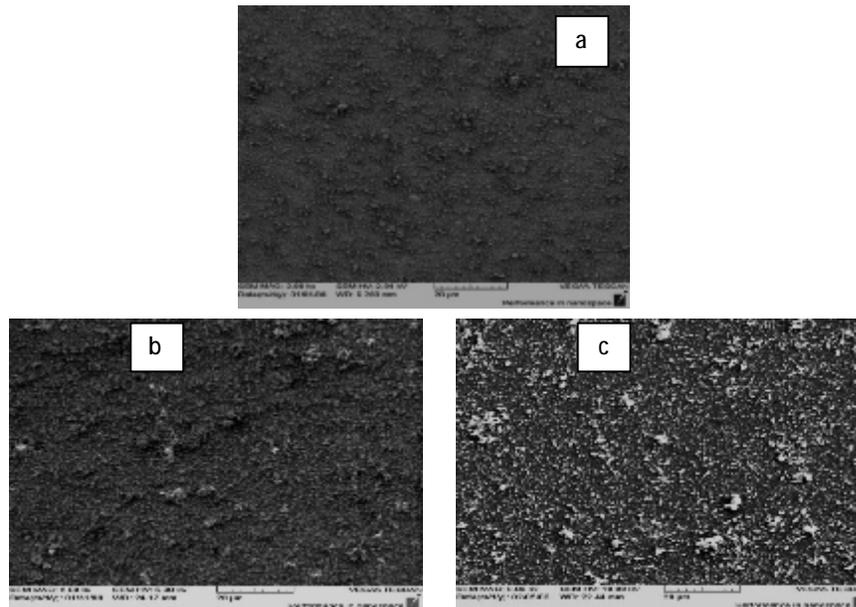


Figure (3) SEM images of (TiO<sub>2</sub>-SnO<sub>2</sub>) films grown on Si at different SnO<sub>2</sub> content% a) 25% b) 50% c) 75.

Figure (4) shows the optical transmittance of the (TiO<sub>2</sub>-SnO<sub>2</sub>) films deposited on glass substrate at oxygen pressures ( $5 \times 10^{-1}$ ) mbar and at fixed substrate temperature of 400°C with 1.4 J/cm<sup>2</sup> laser fluence energy density. With average thickness (200) nm. It is found that the optical transmission of the (TiO<sub>2</sub>-SnO<sub>2</sub>) films increases as SnO<sub>2</sub> content is increased. This may be attributed to the fact that new defects are introduced after Sn atoms substitute Ti atoms and enter into TiO<sub>2</sub> lattice due to the electronegativity and ionic radius difference between Ti and Sn

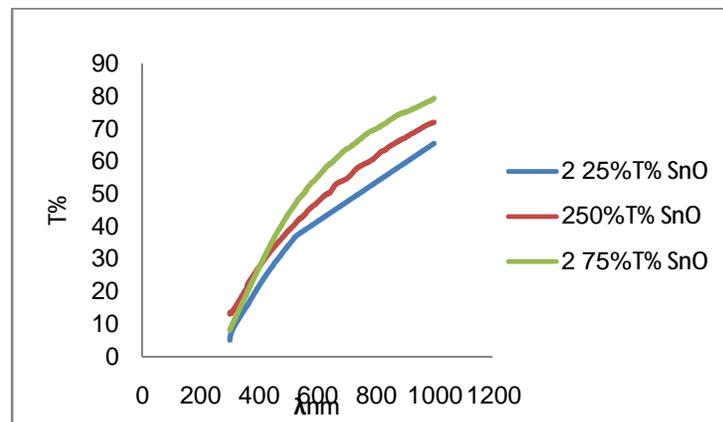
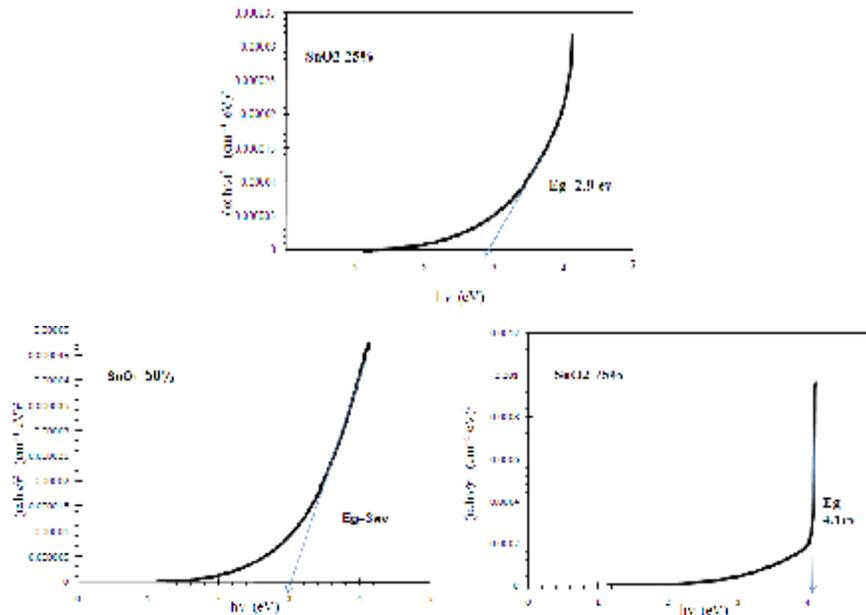


Figure (4) UV-VIS transmittance spectra of the (TiO<sub>2</sub>-SnO<sub>2</sub>) films at different SnO<sub>2</sub> content at 400 °C substrate temperature with laser fluence 1.4J/cm<sup>2</sup>.

The optical energy gap (E<sub>g</sub>) values of the (TiO<sub>2</sub>-SnO<sub>2</sub>) films deposited on glass substrate at constant substrate temperature 400 °C, 1.4 J/cm<sup>2</sup> laser fluence, and oxygen pressure 5×10<sup>-1</sup> mbar are determined and found to be increase from 2.9 to 4.1eV with SnO<sub>2</sub>-content increased as shown in Figure (5). In other words, the optical energy band gap (TiO<sub>2</sub>-SnO<sub>2</sub>) thin films become wider as SnO<sub>2</sub> content increases and The reason for observed blue shift in the band gap could be attributed to the higher band gap energy of SnO<sub>2</sub> (≈ 4.3eV).



Figures (5) A plots of  $(\alpha h\nu)^2$  verses photon energy  $(h\nu)$  of (TiO<sub>2</sub>-SnO<sub>2</sub>) thin films at different SnO<sub>2</sub> content at 400 °C substrate temperature with laser fluence 1.4 J/cm<sup>2</sup>.

### CONCLUSIONS

The (TiO<sub>2</sub>-SnO<sub>2</sub>) mixed oxide thin films have been prepared by PLD with different ratio with the ultimate aim to gain a deeper understanding of the properties of the system.

When the SnO<sub>2</sub> concentration further increased, the XRD analysis did not detect the TiO<sub>2</sub> phase, these due to the small molecular weight of TiO<sub>2</sub>. SEM images show clearly that size and shape of grains are strongly affected by the chemical composition of (TiO<sub>2</sub>-SnO<sub>2</sub>). Were Addition of SnO<sub>2</sub> reduces the grain size to 25.5nm. The band gap energy of (TiO<sub>2</sub>-SnO<sub>2</sub>) films increases as SnO<sub>2</sub> concentration is increased because the higher band gap energy of SnO<sub>2</sub> (≈ 4.3eV).

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